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Magnetically-controllable hollow multi-shell magnetite spheres as stable and reusable SERS substrates



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Yu-Fang Huang^{a,b}, Wen-Chi Huang^a, Pei-Kai Hsu^a, Jenn-Ming Song^{c,d,e,*}, Alexandre Gloter^f, Shih-Yun Chen^{a,*}

^a Department of Materials Science and Engineering, National Taiwan University of Science and Technology, Taipei, Taiwan

^b Material and Chemical Research Laboratories, Industrial Technology Research Institute, Hsinchu, Taiwan

^c Department of Materials Science and Engineering, National Chung Hsing University, Taichung, Taiwan

^d Research Center for Sustainable Energy and Nanotechnology, National Chung Hsing University, Taichung 402, Taiwan

^e Innovation and Development Center of Sustainable Agriculture, National Chung Hsing University, Taichung 402, Taiwan

^f Laboratoire de Physique des Solides, Université Paris-Saclay, CNRS UMR 8502, F-91405 Orsay, France

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ABSTRACT

This study successfully synthesized magnetically-manipulatable and reusable SERS active buoyant substrates, *i.e.* Ag@SiO₂@Fe₃O₄ composite hollow spheres. The composites comprised mesoscopic hollow spheres with a Fe₃O₄ inner shell of 15–30 nm in thickness and strong ferromagnetism, which were synthesized using spray pyrolysis and the subsequent reduction, and a SiO₂ outer shell to maintain the stability of Fe₃O₄, as well as Ag nano-particles on the sphere surface providing localized surface plasmon resonance. The coercive field (H_c) of bare Fe₃O₄ hollow sphere reached 200 Oe, and the saturation magnetization (M_s) was 130 emu/g at 300 K, which was at least 30% higher than the reported values no matter in bulk or nanoscales. Compared to composite spheres with Fe₃O₄ solid cores, the SERS signal intensity of hollow structures was 2.6 times greater, which could be attributed to the buoyancy of hollow structure which contribute to more interaction with the target analytes. Under external magnetic field, these ferromagnetic hollow composites can be concentrated and separated easily, resulting in a more augmented SERS effect (about 1.5 times higher than those collected without applying magnetic field). This study also demonstrated the possibility to reuse those composites for SERS measurements.

1. Introductions

In the last decades, numerous works have been devoted to improve surface enhanced Raman scattering (SERS) detection techniques, in particular, the design and synthesis of SERS substrates. SERS arises from the electromagnetic enhancement, which is induced by local surface plasmon resonance of colloidal or immobilized noble metal nanoparticles (NPs) [1–3]. SERS has been widely used in various fields, including biology, medical pharmacy, materials science, electrochemistry, environmental monitoring, health monitoring, *etc*.

Many methods have been proposed to intensify SERS signals. The application of magnetic materials as a part of SERS substrate is an important one. With the assistance of magnetic matters in SERS substrates, the detection sensitivity could be improved and the substrate recycling becomes approachable, since magnetism can be used to concentrate and collect SERS substrates from the solution [4–7]. Considerable research efforts have been devoted to the synthesis of

functionalized magnetic materials as SERS substrates [8,9], for instance, iron oxides, which have attracted much interest owing to their unique magnetic properties, biocompatibility, and high chemical stability [10]. With Ag or Au decorated on the surface, Fe_2O_3 and Fe_3O_4 in the forms of nanoparticles, nanospheres and dumbbells have been applied as SERS substrates to detect different analytes in liquids [4–6].

Based on the concept of the "Lab on a bubble" [11], hollow spheres have been proposed for SERS substrates. Hollow structures have many advantages, including larger specific surface area, better dispersibility in liquid media, and greater ability of the particles to withstand volumetric cycle changes [12]. It has been reported that both the size and the shape make impact on the magnetic properties of iron oxides [13,14]. Compared to flat surfaces, a higher coercivity and remanence will be observed as in nanobowls and macroporous films, since the curved surfaces pin more domain walls [15,16].

In this study, we develops a stable and reusable LoB structure which exhibit excellent SERS performance, *i.e.* Ag nanoparticle-decorated

* Corresponding authors.

E-mail addresses: samsong@nchu.edu.tw (J.-M. Song), SYChen@mail.ntust.edu.tw (S.-Y. Chen).

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Fig. 1. Magnetically manipulatable 3D SERS substrates developed in this study.

hollow SiO_2/Fe_3O_4 composite spheres. As illustrated in Fig. 1, Fe_3O_4 hollow spheres synthesized using spray pyrolysis are buoyant and magnetically manipulatable. A thin SiO_2 outer shell can isolate Fe_3O_4 from the environment and also eliminate the undesired Raman scattering signal from Fe_3O_4 [17,18,19]. Plasmonic Ag nanoparticles deposited on the sphere surface are able to enhance Raman signals. The phase identification of iron oxides and the composites will be performed by electron energy loss spectrum (EELS), and the magnetic properties will also be investigated. The SERS effect of the hollow composite spheres will be evaluated by Raman spectra of Rhodamine 6G. The influence of hollow structure and SiO_2 coating on SERS activity of Fe_3O_4 spheres will be clarified. Moreover, the reusability of the proposed hollow composite SERS substrates will also be studied.

2. Experimental procedures

2.1. Sample preparation

Hollow iron oxide spheres were prepared using the spray pyrolysis method. Iron (II) chloride (FeCl₂, anhydrous, 99.5% (metals basis), Alfa

Aesar) and Iron (III) chloride (FeCl₃, anhydrous, 98%, Alfa Aesar) were used as precursors, and de-ionized water was the solvent. Glycine (GN, $C_2H_5NO_2$, 98%, Acros Organics) was added to obtain hollow structures. 0.1 M solution was atomized into small droplets using an atomizer (nebulizer, King Ultrasonics Co., Ltd, Taipei, Taiwan) and then sequentially underwent solvent evaporation, solute precipitation and precursor decomposition in the heated tubular reactor at 550 °C in ambient atmosphere. Iron oxide spheres thus formed were collected using a cylindrical electrostatic collector with high voltage potential. A post annealing was performed in Ar/H₂ or N₂/H₂ atmosphere at 350 °C. Using TEOS (Tetraethoxysilane, SiC₈H₂₀O₄) as the starting material, SiO₂ shell was then coated on the iron oxide spheres by sol–gel method. Silver nanoparticles were then deposited on the sphere surface via the incipient wetness method. Silver nitrate (AgNO₃) was the raw material.

2.2. Material characterization

All the samples were characterized by X-ray Diffractometer (XRD) with Cu K radiation (Bruker D2 PHASER). Their morphologies were observed using scanning electron microscope (SEM, JEOL JSM-6500F) and transmission electron microscope (TEM, JEOL JEM-2100F). The oxide spheres were sectioned using ultramicrotomy (LEICA ULTRACUT UC7). The practice is described in detail elsewhere [20]. Cs corrected STEM (JEOL JEM-ARM200F), operated at 200 kV and equipped with electron energy loss spectroscopy (EELS), was applied for phase identification. X-ray absorption measurements were carried out at National Synchrotron Radiation Research Center (NSRRC) in Taiwan. The X-ray absorption near edge fine structures (XANES) spectra at the Ag K-edge was recorded at HSGM beam line 01C and the energy resolution was set to 1.25 eV. Raman scattering was performed with a spectrometer equipped with a micro Raman system (HORIBA iHR550) and a diode laser at an excitation wavelength of 532 nm. The magnetization was measured at room temperature using a vibrating sample magnetometer (VSM).



Fig. 2. (a) XRD and (b) SEM, (c) TEM, (d) cross sectional TEM results, and (e) EELS analysis of iron oxide spheres prepared by spray pyrolysis method.

2.3. Experimental results and discussion

1. Characteristics of iron oxide hollow spheres

The phase of iron oxide spherical particles was determined by X-ray diffraction analysis (Fig. 2(a)). Characteristic diffraction peaks agree with standard α -Fe₂O₃ (JCPDS-0690629). The diffraction peaks verify that the as-synthesized spheres were well-crystalline α -Fe₂O₃. No other phases could be detected. SEM micrographs given in Fig. 2 (b) clearly show that the spheres are with uniform shape, and the sphere size ranged from 300 nm to 1.2 µm. TEM image and corresponding inserted selected-area electron diffraction (SAED) pattern (Fig. 2 (c)) again demonstrate the as-synthesized spheres were α -Fe₂O₃. The cross-sectional TEM images (Fig. 2(d)) indicate that those α -Fe₂O₃ spheres were with continuous and uniform shell, of which the thickness was between 15 and 30 nm. Notably, two kinds of the shell appearance can be distinguished, *i.e.* one with smooth surface and the other one with textured patterns.

EELS analysis was performed to examine the differences between these two kinds of iron oxide shells (Fig. 2(e)). For ferric and ferrous oxides (Fe₂O₃ and FeO), the EELS Fe $L_{2,3}$ edges can be primarily described as transitions from a ground state, Fe $2p^63d^5$ ($2p^63d^6$) to a final excited state, Fe $2p^53d^6$ ($2p^53d^7$), respectively. Fe₂O₃ to FeO can thus be differentiated and the Fe *L* edge of the iron oxide spheres are plotted in Fig. 2 (e) along with hematite and magnetite reference spectra. The energy loss near edge spectrum (ELNES) of the L_3 edge show a clear shoulder at 708 eV, with a maximum located at 710 eV. The spectra obtained from both types of hollow sphere shells (smooth and textures) are found to be the same as the given α -Fe₂O₃ reference [21,22].

It is expected that Fe₃O₄ possesses a larger magnetization than other iron oxides [23]. Accordingly, the as-synthesized α -Fe₂O₃ hollow spheres need to be annealed in various reducing atmospheres for phase transformation. As tabulated in Table 1, samples I, II and III represent iron oxide hollow spheres subjected to post annealing in Ar + H₂ at 350 °C for 1 h, in Ar + H₂ at 350 °C for 2 h, and in N₂ + H₂ at 400 °C for 1 h, respectively. The morphology and phase identification of the sample I are shown in Fig. 3. SEM image (Fig. 3 (a)) and TEM crosssectional observation (Fig. 3(b)) again indicate the uniform shape and hollow structure of the iron oxide spheres subjected to reductive annealing. XRD analysis verifies that the oxide phase turns from α -Fe₂O₃ to Fe₃O₄ or γ -Fe₂O₃ (Fig. 3(c)). However, it is difficult to distinguish Fe₃O₄ and γ -Fe₂O₃ from XRD patterns. Therefore, further EELS analytical results are given as follows.

It has been reported that Fe₃O₄ possesses a mixed valence state and the L_3 edge appears almost shapeless [24–26], while in the case of α -Fe₂O₃ and γ -Fe₂O₃ which are with primary trivalent ground state, a visible splitting in the L_3 can be observed. EELS spectra given in Fig. 3(d) show that in the case of sample I, the Fe *L* edge obtained from spheres with textured surface is consistent with that of the Fe₃O₄ standard. The ELNES of Fe *L* edge obtained from spheres with smooth surface is relatively closer to a ferric edge and could thus be referred to γ -Fe₂O₃. Most of the hollow spheres could be confirmed as Fe₃O₄, and a minor portion of them were γ -Fe₂O₃. Post annealing at the same conditions (Ar + H₂ at 350 °C) but longer time (the sample II) brought about similar results (Fig. S1). Fe₃O₄ hollow spheres are the majority. In

Table 1

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Designation	of sar	nples and	1 their	conditions.

Samples	Conditions
Ι	Iron oxide hollow spheres annealed in Ar $+$ H ₂ at 350 °C for 1 h
II	Iron oxide hollow spheres annealed in Ar $+$ H ₂ at 350 °C for 2 h
III	Iron oxide hollow spheres annealed in N_2 + H_2 at 400 °C for 1 h
IV	Ag@SiO ₂ @Fe ₃ O ₄ hollow spheres with Ag/Fe ₃ O ₄ of 0.71
V	Ag@SiO ₂ @Fe ₃ O ₄ hollow spheres with Ag/Fe ₃ O ₄ of 0.83
VI	$Ag@SiO_2@Fe_3O_4$ hollow spheres with Ag/Fe_3O_4 of 0.91

contrast, for α -Fe₂O₃ hollow spheres annealed in N₂ + H₂ (sample III), the major portion of the spheres were identified as γ -Fe₂O₃ rather than Fe₃O₄ (Fig. S2). This brings us to believe that annealing atmosphere affects the phase transformation of iron oxides.

The magnetization curves of hollow iron oxide spheres under different conditions were measured using VSM at room temperature, and the results are plotted in Fig. 4. In order to clarify the effect of hollow structures, the solid-cored Fe₃O₄ spheres with similar sizes was also prepared for comparison. Fig. 4 illustrates that as-synthesized hollow spheres (α-Fe₂O₃) exhibited a very weak magnetization hysteresis. The coercive field (H_c) was 80 Oe and the saturation magnetization (M_s) was only 10 emu/g. If the saturation magnetization M_s is concerned, the increasing order was hollow α -Fe₂O₃, sample III (71 emu/g), solid cored Fe₃O₄ (73 emu/g), sample II (123 emu/g) and then sample I (130 emu/g). With respect to coercive field (H_c) , the increasing order was α-Fe₂O₃, sample III (150 Oe), solid cored Fe₃O₄ (165 Oe), sample II and then sample I (200 Oe). It is also noticeable that hollow Fe₃O₄ sphere, sample I and sample II, exhibited a tremendous coercive field H_c of about 200 Oe, which was larger than most of the reported values (e.g. nanoparticle (80 Oe), bulk (20 Oe) and hollow spheres [15]). Moreover, samples I and II also possessed a high magnetic saturation (130 emu/g), which was overwhelmingly greater than those in literature (e.g. nanowires (71 emu/g) [27], nanopyramid (52.5 emu/g) [28], nanocubes (40 emu/g) [29], and nanospheres (31 emu/g [29] and 90 emu/g [30]), as well as nanoporous structures (95 emu/g) [31]).

To further understand the unique hysteresis behavior of Fe₃O₄ hollow spheres, the domain state was investigated through a Day plot analysis, where the $M_{\rm rs}$ / $M_{\rm s}$ were plotted as a function of $H_{\rm cr}/H_{\rm c}$ [32]. M_{rs} and H_{cr} are the remanent magnetization and the remanent corrective field respectively, and can be determined experimentally following the method described by Ye et al. [15]. Based on these values, singledomain (SD), pseudosingle-domain (PSD), and multidomain (MD) states of magnetite can be distinguished [33,34]. Ye et al. showed that the Day plots for hollow magnetite spheres appeared in the PSD region, and shifted to the SD region when the temperature was down to 10 K or below [15]. In this study, the behavior of the hollow Fe₃O₄ spheres (sample I) was investigated. The value of M_{rs} and H_{cr} were determined according to the M-H curve measurement under different applied magnetic field, as shown in Fig. S3. The values of $M_{\rm rs}$ / $M_{\rm s}$ and $H_{\rm cr}$ / $H_{\rm c}$ at 300 K were found to be 0.45 and 1.5, respectively, indicating that despite their overall large size, sample I spheres approached the SD region even at 300 K. This could be interpreted in terms of the formation of vortex and onion states theoretically predicted for hollow spheres of magnetically soft ferro-magnet [15,35].

2. Characteristics of Ag@SiO2@Fe3O4 composite hollow spheres

Considering that sample I possessed the highest magnetization, it was selected as the template to prepare SERS substrates. A thin SiO_2 layer and Ag nanoparticles were deposited on sample I in sequence in order to protect isolate Fe_3O_4 and arise SERS effect respectively. TEM observations demonstrate that both concentration of TEOS and synthesis time affect the thickness of SiO_2 layer (Fig. S4). A 12 nm-thick SiO_2 layer was adopted hereafter, which was prepared using 0.4 M TEOS with reaction time of 12 h.

Through the control of the precursor concentration in the synthesis of Ag nanoparticles, the size as well as the distribution of Ag NPs can be determined. As also listed in Table 1, the Ag to Fe_3O_4 mole ratio was set to be 0.71, 0.83, and 0.91. The Ag@SiO₂@Fe₃O₄ composite spheres thus obtained were designated as the samples IV, V, and VI, respectively. Their morphologies are shown in Fig. 5 (a)(b)(c), and the TEM/EDX mapping reveals the elemental distribution (Fig. S5). High resolution lattice image shown in Fig. 5 (d) verifies that the Ag nanoparticles were well crystalline and closely attached on the sphere surface. The lattice fringe spacing of Ag was measured to be 0.236 nm (red line in the inset), matching the d values of the (111) planes of the FCC



Fig. 3. Microstructure investigation of iron oxide hollow spheres after reducing in Ar + H_2 at 350 °C for 1 h. (a) SEM image, (b) cross section images, (c) XRD, and (d) EELS spectrum and the standards.



Fig. 4. Magnetic measurement of iron oxide hollow sphere of different phases.

Ag. For the samples IV, V, and VI, the silver particle diameter was 28 nm, 18 nm, and 27 nm, respectively, while the average Ag nanoparticles spacing was 22 nm, 10 nm, and 11 nm. This suggests that sample V possessed the smallest and closest Ag nanoparticles on oxide sphere surface.

The electronic structure of cations was further investigated by EELS and XAS. ELNES of Fe *L* edge (Fig. 5(e)) and XANES of Fe *K* edge (Fig. 5(f)) obtained from sample V both imply that the iron oxide was not pure Fe₃O₄ after the chemical synthesis of Ag nanoparticles. In Fig. 5 (e), the shoulder on the peak in low energy side might result from partial oxidization of Fe₃O₄ to γ -Fe₂O₃. In Fig. 5 (f), the shift of edge to high energy side also indicates the presence of γ -Fe₂O₃ [36]. The XANES of Ag L edges from the Ag nanoparticles of samples V and the standard spectrum are shown in Fig. 5(g). The configuration of Ag metal is $d^{10}s^1$ and the Ag L edge intensity has been reported to be sensitive to the charge transfer that modulates the filling of d and s bands. With respect to pure Ag, three weak peaks respectively located 2.0, 5.2, and 8.8 eV above the threshold (around 3350 eV) could be observed, which have been assigned to $2p \rightarrow s$ band transitions [37]. As seen in Fig. 5 (g), the fine structures of Ag spectra detected from Ag@ SiO₂@Fe₃O₄ hollow composite sphere and bare Ag NPs were identical. This doesn't agreed to the cases of Ag@CeO₂ [38] and Au-Ag alloys [39], where an increase in the d band population of Ag was observed due to charge transfer. Above spectroscopic investigations of sample V indicate that the covering of SiO₂ layer could prevent charge transfer between Ag nanoparticles and oxide layer, though partial oxidization of Fe₃O₄ was still observed. On the other hand, Ag@Fe₃O₄ hollow composites without a SiO₂ barrier layer, a slightly shift of Fe K edge (Fig. S6 (a)) and a noticeable increase of the Ag L edge intensity was shown, as indicated in red arrows in Fig. S6 (b), implying that both the degree of oxidization of Fe₃O₄ and the charge transfer between Ag and Fe₃O₄ were stronger without the cover of SiO₂ layer.

3. Performance of the Ag@SiO2@Fe3O4 hollow composite spheres

The M-H curves of Fe₃O₄ based hollow composite spheres are given in Fig. 6. The saturation magnetization of Ag@SiO₂@Fe₃O₄ hollow composite spheres were quite similar, in a decreasing order they are sample V, sample VI, and then sample IV. All of them possessed the M_s greater than the one without SiO₂ coating (hollow Ag@Fe₃O₄ in Fig. 6), but apparently lower than those with bare hollow Fe₃O₄ spheres (samples I, II, and III in Fig. 4). The degradation in M_s for composite spheres can be attributed to the attachment of nonmagnetic Ag nanoparticle and amorphous SiO₂ as well, thereby a significant reduction of



Fig. 5. TEM results of Ag@SiO₂@Fe₃O₄ hollow sphere composites with different content of Ag (a) sample IV, (b) sample V, and (c) sample VI. (d) HRTEM of Ag NPs on the surface of sample V. (e) EELS analysis of Fe *L* edge. (f) Fe *K* edge and (g) Ag *L* edge of the iron oxide sphere after Ag deposition.



Fig. 6. Magnetic measurement of $Ag@SiO_2@Fe_3O_4$ hollow sphere with different amounts of Ag. The M-H curve of H-Fe_3O_4@Ag was also plotted.

iron oxide weight fraction. In addition, the partial oxidization of Fe_3O_4 to γ -Fe₂O₃ in the preparation of Ag nanoparticles, as demonstrated by EELS analysis, also leads to the weakening of saturation magnetization. The M_s values for present Ag@SiO₂@Fe₃O₄ hollow composite spheres composite are comparable to the recently reported nanoparticle based magnetic composites (*e.g.* Ag@SiO₂@Fe₃O₄ NPs or Ag@C@Fe₃O₄ NPs), which ranges from 30 to 35 emu/g [40,41].

To evaluate the SERS effect, Ag@SiO₂@Fe₃O₄ hollow composite spheres were used to detect typical analyte Rhodamine 6G (R6G), of which the concentration was set to be 1×10^{-6} M. Ag@SiO₂@Fe₃O₄ composite spheres with Fe₃O₄ solid core were also used for comparison. The SERS spectra of hollow composite spheres are illustrated in Fig. 7 (a), indicating Raman peaks at 1310, 1363, 1509, 1572 and 1650 cm⁻¹, which could be accordance with the reported values. As for hollow composite spheres, sample V presented the most remarkable Raman signals. This can be referred to the combined effects of Ag nanoparticles spacing and size on SERS performance [33]. The signal intensity was stronger than the other two hollow composite spheres, as well as the solid ones. Take the Raman peak at 1650 cm⁻¹ as example, it can be found that the signal intensity of sample V was 2.65 times greater than that of sold composite spheres.

The ability to form well suspension of those hollow SERS spherical substrate can be demonstrated in Fig. 7(b). The dispersion of Ag@ SiO₂@Fe₃O₄ hollow composite spheres in water (indicated by "H") was much better than those with solid core (indicated by "S"), which settled down much faster than the hollow ones. It is interesting that when a magnetic field was applied (the right bottle in Fig. 7(c) and the magnet was placed on the right side), the Ag@SiO2@Fe3O4 hollow composite spheres can be immediately gathered, and completely separated from the solution within only 10 s. The above results suggest that the developed magnetic hollow composite spheres realize well suspension and rapid separation, which can effectively enhance SERS performance. The magnetically concentrated sample V possessed 1650 cm⁻¹ centered Raman peak with the signal intensity around 1.5 times higher than sample V without applying magnetic force, and 3.87 time greater than those with solid cores. The superior SERS performance of hollow sphere sample (e.g. sample V) than the solid spheres can be axcribed to following reasons. Solid spheres and hollow spheres with the same weight were adopted in SERS test, thereby the number of hollow spheres will be greater and thus a higher surface area as well as more Ag nanoparticles on the sphere surface, resulting in better SERS performance. In addition, the buoyancy resulted from hollow structure made the hollow spheres possessed better suspension ability in the solution and thus higher likelihood to react with analytes.

To evaluate reusability of the Ag@SiO2@Fe3O4 hollow composite spheres, the same composites were used to detect again and again. After each measurement, the composites were cleaned with ethanol to remove residual molecules and ions, and separated using an external magnetic field. The renewed composites were then collected for next SERS detection. Fig. 8 shows the recycling test for SERS detection using sample V. Two different dye molecules, R6G and MB (methylene blue), were adopted in the recycling tests. After each detection, the composite was cleaned in ethanol and collected. As shown in Fig. 8 (a) and (b), SERS measurements were also performed every time the sphere substrates were reacted with the analytes (blue spectra) to evaluate the recycling feasibility, and performed when the substrates were just washed before next detection (red spectra) to examine dye molecule residues. It could be found that there were residual dye molecules detected, but the amount was quite limited (the intensity was 100 times less than that before cleaning), and did not affect next recycling test. Remarkably, after eight cycles, it still retained high SERS sensitivity for



Fig. 7. (a) SERS results of $Ag@SiO_2@Fe_3O_4$ hollow sphere composites with different amount of Ag NPs on the surface. (b) Comparison of the suspension in solution between hollow sphere and solid sphere. (c) Illustration of concentration of hollow spheres under external magnetic field.

R6G and MB, and realized reusability. The decrease in the Raman intensity might be attributed to the detachment of Ag nanoparticles from oxide sphere surface during the cleaning process, which was demonstrated by UV–vis spectral analysis (Fig. S7), where the absorption of Ag nanoparticles gradually decreases with increasing recycling numbers.

3. Conclusions

This study proposed and successfully developed Ag@SiO₂@Fe₃O₄ hollow composite spheres exhibiting excellent ferromagnetism and high SERS activity. Fe₃O₄ hollow microspheres were synthesized a mass-production method, spray pyrolysis, and subsequent post annealing to achieve α -Fe₂O₃ to Fe₃O₄ transformation. The saturation magnetization of the Fe₃O₄ hollow microspheres can be up to 130 emu/g at 300 K, which was much higher than the reported values. With a protective SiO₂ coating and Ag nanoparticles providing localized surface plasmon resonance, it has been demonstrated that Ag@SiO₂@Fe₃O₄ hollow composite spheres can be well suspended and manipulated with the application of an external magnetic field. Both these two characteristics

effectively enhance SERS performance. The reusability of the hollow composite SERS substrates was also demonstrated. The SERS signals were still distinguishable even in the 8th test.

CRediT authorship contribution statement

Yu-Fang Huang: Formal analysis, Validation. Wen-Chi Huang: Formal analysis, Validation. Pei-Kai Hsu: Formal analysis, Validation. Jenn-Ming Song: Conceptualization, Investigation, Resources, Writing - review & editing. Alexandre Gloter: Methodology, Resources, Writing - review & editing. Shih-Yun Chen: Investigation, Conceptualization, Formal analysis, Writing - original draft, Writing review & editing, Supervision, Funding acquisition, Project administration.

Declaration of Competing Interest

None.



Fig. 8. SERS spectra of the reproducible measurements for the $Ag@SiO_2@Fe_3O_4$ hollow sphere substrate (sample V). Blue spectra were detected when the sphere substrates were reacted with the analytes, and red spectra were measured when the substrates were just cleaned by ethanol between two recycling tests. Two different dyes were used: (a) R6G and (b) MB. The concentration was 10^{-6} M. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

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Appendix A. Supplementary material

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